# **pH-Dependent Structure and Properties of TiO2/SiO2 Nanoparticle Multilayer Thin Films**

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We demonstrate that the structure and properties of layer-by-layer assembled multilayer thin films comprising positively charged  $TiO<sub>2</sub>$  and negatively charged  $SiO<sub>2</sub>$  nanoparticles can be varied by controlling assembly conditions. The average incremental thickness of the all-nanoparticle thin films after a pair of  $7 \text{ nm}$  TiO<sub>2</sub> and 22 nm SiO<sub>2</sub> nanoparticle adsorption steps was strongly dependent on the pH of each nanoparticle suspension, varying from a few nanometers to 30 nm. The surface charge density of the adsorbing nanoparticles and that of the previously adsorbed nanoparticle layers are the crucial factors in determining the average bilayer thickness of the films. Zeta-potential measurements of polystyrene microspheres coated with TiO<sub>2</sub>/SiO<sub>2</sub> nanoparticle multilayers indicated that incomplete charge reversal after the deposition of nanoparticles leads to a very small thickness increase at each adsorption step, whereas complete charge reversal results in significantly larger thickness increments. Other film characteristics including refractive index, porosity, and chemical composition of the multilayers also depended on assembly conditions but to a much lesser extent compared to the dependence observed for average bilayer thickness.

#### **Introduction**

The formation of thin films containing nanoparticles can lead to novel applications in the areas of photonics, catalysis, electronics, magnetism, and biomedical engineering.<sup>1</sup> Despite the recent advances in creating various types of functional nanoparticles, $2^{-4}$  it still remains a challenge to create conformal thin films of nanoparticles with precise control over the physicochemical properties. One approach that has been widely used is the layer-by-layer (LbL) deposition<sup>5</sup> of charged nanoparticles in conjunction with an oppositely charged polyelectrolyte. $6-16$  Properties of nanoparticle-

- (1) Fendler, J. H. *Nanoparticles and Nanostructured Films: Preparation, Characterization and Applications*; Wiley-VCH Verlag GmbH: Weinheim, Germany, 1998.
- (2) Alivisatos, A. P. *Science* **<sup>1996</sup>**, *<sup>271</sup>*, 933-937.
- (3) Hyeon, T. *Chem. Commun.* **<sup>2003</sup>**, 927-934.
- (4) Pellegrino, T.; Kudera, S.; Liedl, T.; Javier, A. M.; Manna, L.; Parak,
- W. J. *Small* **<sup>2005</sup>**, *<sup>1</sup>*, 48-63. (5) Decher, G.; Schlenoff, J. B., Eds. *Multilayer Thin Films: Sequential Assembly of Nanocomposite Materials*; Wiley-VCH Verlag GmbH & Co.: Weinheim, Germany, 2003.
- (6) Lvov, Y.; Ariga, K.; Onda, M.; Ichinose, I.; Kunitake, T. *Langmuir* **<sup>1997</sup>**, *<sup>13</sup>*, 6195-6203.
- (7) Cebeci, F. C.; Wu, Z. Z.; Zhai, L.; Cohen, R. E.; Rubner, M. F. *Langmuir* **<sup>2006</sup>**, *<sup>22</sup>*, 2856-2862.
- (8) Caruso, R. A.; Susha, A.; Caruso, F. *Chem. Mater.* **<sup>2001</sup>**, *<sup>13</sup>*, 400- 409.
- (9) Ai, H.; Jones, S. A.; Lvov, Y. M. *Cell Biochem. Biophys.* **2003**, *39*,  $23 - 43$ .
- (10) Cant, N. E.; Zhang, H. L.; Critchley, K.; Mykhalyk, T. A.; Davies, G. R.; Evans, S. D. J. Phys. Chem. B 2003, 107, 13557-13562. G. R.; Evans, S. D. *J. Phys. Chem. B* **<sup>2003</sup>**, *<sup>107</sup>*, 13557-13562.
- (11) Liang, Z. Q.; Dzienis, K. L.; Xu, J.; Wang, Q. *Ad*V*. Funct. Mater.* **<sup>2006</sup>**, *<sup>16</sup>*, 542-548.

containing multilayers can be precisely tuned by controlling the assembly conditions (e.g., salt concentration and pH conditions) and the types of nanoparticles and polyelectrolytes assembled. Nanoparticle-containing multilayers have been studied extensively for their potential use in the various fields of science<sup>5</sup> and have recently shown promise in creating thin films with extreme wetting behaviors.<sup>7,17,18</sup> Also, these nanoparticle-containing multilayers can be readily coated onto nonplanar geometries such as spherical colloidal particles and porous membranes.<sup>8,13,19-21</sup>

Although researchers in this field have generally shown that multilayer thin films comprising nanoparticles and oppositely charged polyelectrolytes can be readily prepared, only a few studies have shown that multilayers of oppositely charged nanoparticles can be assembled.<sup>22,23</sup> It is interesting to note, however, that the very first report of the preparation

- (13) Lu, Z. H.; Prouty, M. D.; Guo, Z. H.; Golub, V. O.; Kumar, C. S. S. R.; Lvov, Y. M. *Langmuir* **<sup>2005</sup>**, *<sup>21</sup>*, 2042-2050.
- (14) Ostrander, J. W.; Mamedov, A. A.; Kotov, N. A. *J. Am. Chem. Soc.* **<sup>2001</sup>**, *<sup>123</sup>*, 1101-1110.
- (15) Rouse, J. H.; Ferguson, G. S. *J. Am. Chem. Soc.* **<sup>2003</sup>**, *<sup>125</sup>*, 15529- 15536.
- (16) Liz-Marzan, L. M.; Mulvaney, P. *J. Phys. Chem. B* **<sup>2003</sup>**, *<sup>107</sup>*, 7312- 7326.
- (17) Kommireddy, D. S.; Patel, A. A.; Shutava, T. G.; Mills, D. K.; Lvov, Y. M. *J. Nanosci. Nanotechnol.* **<sup>2005</sup>**, *<sup>5</sup>*, 1081-1087.
- (18) Zhai, L.; Cebeci, F. C.; Cohen, R. E.; Rubner, M. F. *Nano Lett.* **2004**, *<sup>4</sup>*, 1349-1353.
- (19) Donath, E.; Sukhorukov, G. B.; Caruso, F.; Davis, S. A.; Möhwald, H. *Angew. Chem., Int. Ed.* **<sup>1998</sup>**, *<sup>37</sup>*, 2202-2205.
- (20) Lee, D.; Nolte, A. J.; Kunz, A. L.; Rubner, M. F.; Cohen, R. E. *J. Am. Chem. Soc.* **<sup>2006</sup>**, *<sup>128</sup>*, 8521-8529.
- (21) Hollman, A. M.; Bhattacharyya, D. *Langmuir* **<sup>2004</sup>**, *<sup>20</sup>*, 5418-5424.
- (22) Kumar, A.; Mandale, A. B.; Sastry, M. *Langmuir* **<sup>2000</sup>**, *<sup>16</sup>*, 6921- 6926.

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<sup>(12)</sup> Krasteva, N.; Besnard, I.; Guse, B.; Bauer, R. E.; Mullen, K.; Yasuda, A.; Vossmeyer, T. *Nano Lett.* **<sup>2002</sup>**, *<sup>2</sup>*, 551-555.

of multilayers involved the layer-by-layer (LbL) deposition of oppositely charged nanoparticles without any organic processing agents.24 This pioneering work did not receive substantial follow-up attention because the robust assembly conditions were not easily achieved for the commonly available nanoparticle systems. Thus, researchers turned their attention to the more readily processed polyelectrolyte containing multilayer systems.5 Recently, we have reexamined the possibility of using the LbL technique in the absence of any polymer components and demonstrated that all-nanoparticle thin film coatings comprising positively charged  $TiO<sub>2</sub>$  nanoparticles and negatively charged  $SiO<sub>2</sub>$ nanoparticles can be readily assembled.<sup>25</sup> These all-nanoparticle multilayer coatings exhibited potentially useful antifogging, antireflection, and self-cleaning properties. The ability to tune various film characteristics, such as chemical composition, would enable the fabrication of functional coatings for specific applications. For example, others have shown that the amount of  $TiO<sub>2</sub>$  nanoparticles in various nanocomposite thin films is crucial in determining their photocatalytic or superhydrophilic properties.<sup>26,27</sup> In this respect, a comprehensive understanding of the effect of assembly conditions on the chemical composition would provide an important blueprint for the fabrication of functional thin films with desirable properties.

Iler in his seminal report describing multilayers of oppositely charged colloidal particles noted that "adsorption of colloidal silica onto positively charged alumina-coated glass substrates was most rapid and complete in the low pH range of 2 to 4".<sup>24</sup> Although a quantitative study on the effect of assembly conditions (i.e., pH of each nanoparticle suspension) was not performed at the time, he clearly noticed that the pH of colloidal particle suspensions had an enormous influence on the deposition characteristics of multilayers. In this study, we show that average incremental bilayer thickness, refractive index, chemical composition, and porosity of all-nanoparticle multilayer thin films comprising  $TiO<sub>2</sub>$  and  $SiO<sub>2</sub>$  nanoparticles can be varied by changing the pH of the nanoparticle suspensions. By independently changing the pH of each nanoparticle suspension, the average bilayer thickness28 of all-nanoparticle thin films could be varied between a few nanometers to tens of nanometers. It will also be demonstrated that the chemical composition as well as the porosity of the all-nanoparticle thin films are dependent on the assembly conditions. Zeta-potential measurements of polystyrene microspheres coated with  $TiO<sub>2</sub>$  and  $SiO<sub>2</sub>$  nanoparticles show that conditions that enable complete charge reversal after the deposition of each nanoparticle layer are

- (24) Iler, R. K. *J. Colloid Interface Sci.* **<sup>1966</sup>**, *<sup>21</sup>*, 569-594.
- (25) Lee, D.; Rubner, M. F.; Cohen, R. E. *Nano Lett.* **<sup>2006</sup>**, *<sup>6</sup>*, 2305- 2312.

- (27) Machida, M.; Norimoto, K.; Watanabe, T.; Hashimoto, K.; Fujishima, A. J. Mater. Sci. 1999,  $34$ , 2569-2574. A. *J. Mater. Sci.* **<sup>1999</sup>**, *<sup>34</sup>*, 2569-2574. (28) The term "average bilayer thickness" is used to represent the
- incremental thickness measured after a sequential exposure of substrates to  $TiO<sub>2</sub>$  and  $SiO<sub>2</sub>$  nanoparticle suspensions. The term "bilayer" in this sense is not meant to indicate that one dense layer of  $SiO<sub>2</sub>$  nanoparticles is formed on top of a single layer of  $TiO<sub>2</sub>$ nanoparticles.

essential to create thick all-nanoparticle multilayers. Incomplete charge reversal leads to sparse nanoparticle adsorption at each deposition step, resulting in a relatively small incremental thickness per bilayer.

#### **Experimental Section**

**Materials.** Anatase titanium oxide nanoparticles STS-100 (18 wt %  $TiO<sub>2</sub>$  suspension in water, average particle size 7 nm, and specific surface area 320 m<sup>2</sup>/g) were generously provided by Ishihara Sangyo Kaisha, Ltd (Japan). Silica nanoparticles Ludox TM-40 (40 wt %  $SiO<sub>2</sub>$  suspension in water, average particle size 22 nm, and specific surface area 140  $m^2/g$ ) were purchased from Sigma-Aldrich (St. Louis, MO). The average size of nanoparticles was provided by the suppliers.<sup>29</sup> Glass slides were purchased from VWR International, and 1.5 *µ*m poly(styrene) (PS) microspheres were purchased from Polysciences, Inc.

**Assembly of TiO2/SiO2 Nanoparticle Multilayers.** Sequential adsorption of  $TiO<sub>2</sub>$  and  $SiO<sub>2</sub>$  nanoparticles onto glass substrates was performed using an automated dipping machine. The concentration of each nanoparticle suspension was adjusted to 0.03 wt %. The pH of each nanoparticle suspension was adjusted using 1.0 M HCl or NaOH. Glass substrates were degreased with 2 vol % detergent solution and then cleaned with 1.0 M NaOH solutions under sonication for 15 min. Deionized water (>18 M $\Omega$  m, Millipore Milli-Q (MQ)) was used to make the nanoparticle suspensions. The dipping time in each nanoparticle solution was 10 min followed by three rinse steps (2, 1, and 1 min) in deionized water. The pH of the nanoparticle suspensions was readjusted after deposition of every three bilayers corresponding to approximately 1.5 h. A Barnstead Thermolyne 47900 furnace was used to calcinate the films at 550 °C for 2 h.

**Characterization.** Thicknesses of the  $TiO<sub>2</sub>/SiO<sub>2</sub>$  nanoparticlebased multilayers assembled on glass substrates were determined using a Woolham Co. VASE spectroscopic ellipsometer; the data analysis was done using the WVASE32 software package. An uncoated glass slide was first scanned in the ellipsometer. Measurements were performed from 300 to 900 nm at an angle of incidence of 70°. The data obtained were fitted to a Cauchy model, which assumes that the real part of the refractive index  $(n_f)$  can be described by

$$
n_{\rm f}(\lambda) = A_{\rm n} + \frac{B_{\rm n}}{\lambda^2} + \frac{C_{\rm n}}{\lambda^4}
$$
 (1)

where  $A_n$ ,  $B_n$ , and  $C_n$  are constants and  $\lambda$  is the wavelength of the light. The values of the refractive index reported in this study were determined at 633 nm.

The zeta-potential and hydrodynamic diameter of the each nanoparticle (or aggregates of nanoparticles) as a function of pH were determined using a ZetaPals (Brookhaven Instrument Corp.) and a 90Plus Particle Size Analyzer (Brookhaven Instrument Corp.), respectively. Measurements were made within 1 h of pH adjustments.

For measurement of zeta-potential after each deposition step,  $TiO<sub>2</sub>/SiO<sub>2</sub>$  multilayers were assembled onto polystyrene (PS) microspheres whose diameter was 1.5 *µ*m. To facilitate the adsorption of nanoparticles, we primed PS microspheres with five

<sup>(23)</sup> Hao, E. C.; Yang, B.; Zhang, J. H.; Zhang, X.; Sun, J. Q.; Shen, S. C. *J. Mater. Chem.* **<sup>1998</sup>**, *<sup>8</sup>*, 1327-1328.

<sup>(26)</sup> Nakajima, A.; Hashimoto, K.; Watanabe, T.; Takai, K.; Yamauchi, G.; Fujishima, A. *Langmuir* **<sup>2000</sup>**, *<sup>16</sup>*, 7044-7047.

<sup>(29)</sup> Average particle sizes of  $SiO<sub>2</sub>$  and  $TiO<sub>2</sub>$  nanoparticles determined by dynamic light scattering were  $24.2 \pm 4.2$  and  $8.7 \pm 3.2$  nm, respectively. Similar values were obtained via transmission electron microscopy; that is,  $24.2 \pm 2.7$  nm for SiO<sub>2</sub> and  $6.5 \pm 1.3$  nm for TiO2 nanoparticles. To be consistent with our previous papers (refs 7 and 25), the nominal size of each nanoparticle as indicated by the suppliers will be used in this work.

bilayers of poly(allylamine hydrochloride) (PAH) and poly(styrene sulfonate) (PSS) (the pH of each polyelectrolyte solution was ∼4, and 0.1 M NaCl was added to each solution) prior to the deposition of  $TiO<sub>2</sub>/SiO<sub>2</sub>$  nanoparticle multilayers. The zeta-potential of the coated microspheres after each adsorption step was measured in pH adjusted deionized water. To create hollow microcapsules of TiO2 and SiO2 nanoparticles, PS microspheres coated with 5 bilayers of TiO<sub>2</sub> and SiO<sub>2</sub> nanoparticles were calcinated at 550 °C for 2 h. These calcinated hollow microcapsules were observed with a JEOL 6320FV field-emission high-resolution scanning electron microscope (SEM) at an acceleration voltage of 2 kV. The samples were coated with 10 nm of Au/Pd prior to SEM observation.

The porosity and chemical composition of the multilayers were determined using a recently reported method<sup>25</sup> based on ellipsometry. In brief, all-nanoparticle multilayers (12 bilayers) assembled on Si substrates at different assembly conditions were first calcinated at 550 °C for 2 h. The calcination prevents the swelling of films in water. The refractive indices of the films were then measured in air  $(n_{f,1})$  and in water  $(n_{f,2})$  using the ellipsometer. The porosity and the chemical composition of the multilayers are then calculated by the following equations

$$
p = \frac{n_{\text{f},2} - n_{\text{f},1}}{n_{\text{f,water}} - n_{\text{f,air}}} = \frac{n_{\text{f},2} - n_{\text{f},1}}{0.33} \tag{2}
$$

$$
v_{\text{TiO}_2} = \frac{n_{\text{f,framework}} - n_{\text{f,SiO}_2}}{n_{\text{f,TiO}_2} - n_{\text{f,SiO}_2}} (1 - p)
$$
 (3)

$$
v_{\rm SiO_2} = 1 - (p + v_{\rm TiO_2})
$$
 (4)

where  $v$ ,  $p$ , and  $n_f$  represent the volume fraction of each nanoparticle, the porosity, and the refractive index of the nanoparticles, respectively. Framework denotes the solid materials in the multilayer thin films that consist of  $TiO<sub>2</sub>$  and  $SiO<sub>2</sub>$  nanoparticles. For detailed derivation of eqs 2-4, see ref 25. Refractive indices of the  $TiO<sub>2</sub>$  and  $SiO<sub>2</sub>$  nanoparticles were determined to be 2.29 and 1.45, respectively. The chemical composition of the multilayers is reported in weight fraction of  $TiO<sub>2</sub>$  nanoparticles using the density of each nanoparticle (TiO<sub>2</sub> = 3.9 g/cm<sup>3</sup> and SiO<sub>2</sub> = 1.3 g/cm<sup>3</sup>).<sup>25</sup>

#### **Results and Discussion**

All-nanoparticle multilayer thin films comprising 7 nm  $TiO<sub>2</sub>$  and 22 nm  $SiO<sub>2</sub>$  nanoparticles were assembled onto glass or Si substrates to determine average bilayer thicknesses.28 Figure 1 is an example of the variation in average bilayer thickness with assembly conditions. In this case, the  $pH$  of the TiO<sub>2</sub> nanoparticle suspension was held constant at pH 3.0, whereas that of the  $SiO<sub>2</sub>$  nanoparticle suspension was varied from 2.0 to 5.0 by a single pH unit. Thicknesses were measured every three bilayers up to 12 bilayers for each condition and then the average bilayer thickness was determined using linear regression. To eliminate the effect of substrates, the regression lines were drawn without forcing the fit through the origin. In all cases, the R-squared values were greater than 0.95, indicating that the multilayers assembled at all conditions showed linear growth behavior up to 12 bilayers. The fact that some of the regression lines do not go through the origin, however, indicates the existence of substrate effects in the first few bilayers of the multilayer films.

Following the procedure described above, 12 different combinations of pH conditions were used to create a matrix



**Figure 1.** Increase in the thickness of  $TiO<sub>2</sub> 3.0/SiO<sub>2</sub> y$  multilayers as a function of the number of deposited bilayers, where  $y$  is the pH of the  $SiO<sub>2</sub>$ nanoparticle suspension. The values of *y* are shown in the legend of the graph. The nanoparticle multilayer thin films were assembled on glass substrates.



**Figure 2.** pH matrix showing the average bilayer thickness of  $7 \text{ nm } TiO_2/$  $22 \text{ nm SiO}_2$  nanoparticle multilayers as a function of nanoparticle suspension pH. At least two measurements were made and averaged for each multilayer. Error in the average bilayer thickness for each multilayer is less than 2.5 nm.

of average bilayer thickness as shown in Figure 2. Depending on the assembly conditions, multilayers can show either very little growth (∼2 nm/bilayer) or a large growth (∼30 nm/ bilayer) after a sequential exposure of substrates to  $TiO<sub>2</sub>$  and SiO2 nanoparticle suspensions. Clear trends are observed as the assembly conditions are changed. For a constant pH of the  $SiO<sub>2</sub>$  nanoparticle suspension, the average bilayer thickness increases as the  $pH$  of the TiO<sub>2</sub> suspension is increased, exhibiting the greatest thickness at pH 4.0. For a constant pH of the TiO<sub>2</sub> nanoparticle suspension, on the other hand, the average bilayer thickness goes through a maximum at  $pH$  3.0 for the  $SiO<sub>2</sub>$  nanoparticle suspensions. Iler also observed that the multilayer assembly of  $SiO<sub>2</sub>$  nanoparticles and positively charged colloidal particles such as alumina showed the maximum growth at pH 3.0 when the pH of the positively charged particle suspension was held constant at 4.0.24 All multilayers shown in Figure 2 yielded completely transparent films with high uniformity. When the pH of the  $TiO<sub>2</sub>$  nanoparticles was increased to pH 5.0, the films became cloudy, independent of the  $pH$  of the  $SiO<sub>2</sub>$  suspensions. As will be described below, the opaqueness of the films assembled at pH 5.0 is likely due to the formation (and



**Figure 3.** (a) Zeta-potential and (b) effective diameter of suspended nanoparticles determined by dynamic light scattering as a function of nanoparticle suspension pH.

adsorption) of large  $TiO<sub>2</sub>$  nanoparticle aggregates in the suspension.

The results obtained in Figure 2 indicate that there exists a narrow regime of pH conditions in which assembly proceeds in a "successful" layer-by-layer manner. Specifically, the average bilayer thickness observed for all conditions except the pH range between 3.0 and 4.0 for both nanoparticle suspensions is significantly less than the diameter of the  $SiO<sub>2</sub>$  nanoparticles themselves. Therefore, successful LbL assembly leading to significant adsorption of nanoparticles at each deposition step occurs only within this narrow assembly window. As will be shown below, the complete charge reversal of multilayers after each nanoparticle deposition step is essential for successful LbL assembly.

To better understand the effect of pH on the growth behavior of the  $TiO<sub>2</sub>/SiO<sub>2</sub>$  nanoparticle multilayers, the zetapotential and the size of nanoparticles (or that of nanoparticle aggregates) as a function of suspension pH were determined as shown in Figure 3. For  $SiO<sub>2</sub>$  nanoparticles, the surface charge density decreases as the pH of the suspension decreases; the isoelectric point is reached around pH 2.0. The size of the  $SiO<sub>2</sub>$  entities in suspension, determined by dynamic light scattering (DLS), does not vary significantly over the pH range that was used for multilayer assembly, indicating that even though the surface charge density of the  $SiO<sub>2</sub>$  particles changes significantly from pH 5.0 to 2.0, the particles are quite stable against flocculation. Excellent colloidal stability of  $SiO<sub>2</sub>$  nanoparticles near their isoelectric point has been well-documented and reported by others.<sup>30-32</sup> In the case of  $TiO<sub>2</sub>$  nanoparticles, the charge density of the nanoparticles continually decreases above pH 3.0. The  $TiO<sub>2</sub>$ nanoparticles used in this study exhibited an isoelectric point near pH 7.0, which is consistent with reported values for  $TiO<sub>2</sub>$  nanoparticles.<sup>33-35</sup> The observed decrease in the zetapotential at pH 2.0 is likely due to compression of the electric double layer as the ionic strength increases at such a low pH.<sup>36</sup> The size of suspended  $TiO<sub>2</sub>$  entities, determined via

- (30) Depasse, J.; Watillon, A. *J. Colloid Interface Sci.* **<sup>1970</sup>**, *<sup>33</sup>*, 430- 438.<br>(31) Depasse, J. J. Colloid Interface Sci. 1999, 220, 174–176.
- (31) Depasse, J. *J. Colloid Interface Sci.* **<sup>1999</sup>**, *<sup>220</sup>*, 174-176.
- (32) Binks, B. P.; Lumsdon, S. O. *Phys. Chem. Chem. Phys.* **<sup>1999</sup>**, *<sup>1</sup>*, 3007- 3016.
- (33) Mandzy, N.; Grulke, E.; Druffel, T. *Powder Technol.* **<sup>2005</sup>**, *<sup>160</sup>*, 121- 126.

DLS, increases with increasing pH, indicating that these nanoparticles flocculate into large clusters at pH values higher than 3.0. At pH 5.0, the size of the aggregates is larger than the  $TiO<sub>2</sub>$  nanoparticles at pH 3.0 by a factor of 5, and some precipitation of nanoparticle aggregates was observed. Our observation of a relatively low colloidal stability of TiO<sub>2</sub> nanoparticles compared to  $SiO<sub>2</sub>$  nanoparticles near the isoelectric point also has been observed by others.  $34,37,38$  As briefly mentioned above, the formation and adsorption of large aggregates of  $TiO<sub>2</sub>$  nanoparticles at pH 5.0 leads to the formation of multilayers that scatter light.<sup>39</sup>

Figure 3a explains the reason behind the observed negligible multilayer growth when the pH of the  $SiO<sub>2</sub>$ nanoparticle suspension is 2.0 regardless of  $TiO<sub>2</sub>$  pH (Figure 2). Because the  $SiO<sub>2</sub>$  nanoparticles are barely charged near their isoelectric point at pH 2.0, their electrostatic interaction with positively charged  $TiO<sub>2</sub>$  nanoparticles is extremely weak, leading to the negligible bilayer growth observed. It is reasonable to postulate that negligible multilayer growth would also be observed if the  $pH$  of the TiO<sub>2</sub> nanoparticle suspension was adjusted to 7.0; however, due to the aggregation and precipitation of  $TiO<sub>2</sub>$  nanoparticles above pH 5.0 (Figure 3b), this regime cannot be accessed experimentally.

The zeta-potential of  $TiO<sub>2</sub>/SiO<sub>2</sub>$  nanoparticle multilayer coated polystyrene microspheres was monitored after each deposition step to provide important insight to the effect of assembly conditions on the surface charge density of multilayers during the adsorption of nanoparticles. For this purpose, two conditions were chosen:  $TiO<sub>2</sub> 3.0/SiO<sub>2</sub> 3.0$  and  $TiO<sub>2</sub> 3.0/SiO<sub>2</sub> 5.0.<sup>40</sup>$  Whereas the former condition exhibited a bilayer growth of ca. 20 nm, the latter showed very limited growth of ca. 2 nm per bilayer. The comparison of these

- (37) Imae, T.; Muto, K.; Ikeda, S. *Colloid Polym. Sci.* **<sup>1991</sup>**, *<sup>269</sup>*, 43-48.
- (38) Grishchenko, L. I.; Medvedkova, N. G.; Nazarov, V. V.; Frolov, Y.
- G. *Colloid J.* **1994**, *56*, 215–217.<br>(39) Average bilayer thicknesses of (TiO<sub>2</sub> 5.0/SiO<sub>2</sub> 3.0) and (TiO<sub>2</sub> 5.0/ SiO<sub>2</sub> 5.0) were determined to be ∼30 and 33 nm, respectively, as determined via ellipsometry. These films are opaque and scatter light.

<sup>(34)</sup> Tkachenko, N. H.; Yaremko, Z. M.; Bellmann, C.; Soltys, M. M. *J. Colloid Interface Sci.* **<sup>2006</sup>**, *<sup>299</sup>*, 686-695.

<sup>(35)</sup> Zhao, J. C.; Hidaka, H.; Takamura, A.; Pelizzetti, E.; Serpone, N. *Langmuir* **<sup>1993</sup>**, *<sup>9</sup>*, 1646-1650. (36) Hunter, R. J. *Zeta Potential in Colloid Science: Principles and*

*Applications*; Academic Press: London, 1981.



**Figure 4.** Zeta-potential variation of (TiO2 3.0/SiO2 3.0) and (TiO2 3.0/SiO2 5.0) multilayers deposited on 1.5 *µ*m polystyrene (PS) microparticles. The PS particles were coated with 5 bilayers of (PAH/PSS) multilayers prior to the deposition of TiO<sub>2</sub>/SiO<sub>2</sub> multilayers. Filled and open symbols in each figure represent zeta-potentials of coated PS microspheres after adsorption of  $SiO<sub>2</sub>$  and  $TiO<sub>2</sub>$  nanoparticles, respectively.

two extreme cases revealed important information regarding the dependence of layer thickness on assembly conditions. After the adsorption of each nanoparticle layer, coated PS microspheres were dispersed in deionized water whose pH was adjusted to the same value as that of the next nanoparticle suspension. For example, in the case of  $TiO<sub>2</sub> 3.0/SiO<sub>2</sub>$ 5.0 multilayers, after the adsorption of  $TiO<sub>2</sub>$  nanoparticles at pH 3.0 and three rinse steps, the coated PS microspheres were then dispersed in pH 5.0 adjusted-DI water for the measurement of zeta-potential. This scheme allowed us to probe the surface charge density at each nanoparticle deposition step and its subsequent effect on the adsorption of nanoparticles.

Figure 4a clearly shows that for  $TiO<sub>2</sub> 3.0/SiO<sub>2</sub> 3.0$ multilayers, after deposition of each nanoparticle layer, the surface charge of the multilayer is completely reversed. This charge reversal leads to a complete adsorption from the next nanoparticle suspension in the processing scheme. It is a wellunderstood fact that charge reversal after deposition of each layer of polyelectrolyte or charged nanoparticles is essential for a successful build-up of multilayer thin films. $41-43$ 

In the case of the  $TiO<sub>2</sub> 3.0/SiO<sub>2</sub> 5.0$  multilayers, however, each nanoparticle adsorption step does not necessarily induce charge reversal (Figure 4b). Specifically, after the deposition step of positively charged  $TiO<sub>2</sub>$  nanoparticles at pH 3.0, the coated-PS microspheres still exhibit a negative surface charge at pH 5.0, whereas the coated-PS microspheres have positive surface charge in pH 3.0 water after the deposition step of silica nanoparticles at pH 5.0. These results preclude any significant electrostatically driven nanoparticle adsorption in this set of processing conditions, consistent with the very low value of bilayer thickness observed at  $TiO<sub>2</sub> 3.0/SiO<sub>2</sub>$ 5.0 (Figure 2). We believe the major reason for the observed negligible growth of multilayers at numerous other conditions

(i.e., conditions except for pH ranges between 3.0 and 4.0) observed in Figure 2 is due to the incomplete charge reversal after the deposition of each nanoparticle layer. Also, it can be concluded that a high surface charge density of suspended nanoparticles (e.g.,  $pH$  5.0 for  $SiO<sub>2</sub>$  nanoparticles) does not necessarily result in a large average bilayer thickness.

The observed incomplete charge reversal is likely due to the changes in the surface charge density of nanoparticles near the surface of the multilayer thin films. For example, when  $SiO<sub>2</sub>$  nanoparticles are adsorbed at pH 5.0, their surface charge density is relatively high, as evidenced by the zetapotential value in Figure 2.  $SiO<sub>2</sub>$  nanoparticles will adsorb on top of previously adsorbed  $TiO<sub>2</sub>$  nanoparticles but because of the high charge density of the  $SiO<sub>2</sub>$  nanoparticles and low surface charge of the previously adsorbed  $TiO<sub>2</sub>$  nanoparticle layers,  $SiO<sub>2</sub>$  nanoparticles cannot adsorb into dense layers on top of the TiO<sub>2</sub> nanoparticle layers. When the  $SiO<sub>2</sub>$ nanoparticle-coated surface is exposed to a  $TiO<sub>2</sub>$  nanoparticle suspension at pH 3.0, the surface charge density of the outermost  $SiO<sub>2</sub>$  nanoparticle layer decreases and that of the underlying  $TiO<sub>2</sub>$  nanoparticles increases. Due to the changes in the surface charge density of nanoparticles near the surface, the multilayer exhibits positive surface charge overall, leading to little adsorption of positively charged  $TiO<sub>2</sub>$ nanoparticles from the suspension. Therefore, continuous exposure of the substrates to nanoparticle suspensions at this condition (i.e.,  $TiO<sub>2</sub> 3.0/SiO<sub>2</sub> 5.0$ ) leads only to the negligible growth observed in Figure 2.

The effect of assembly conditions on the growth of multilayers could be directly observed by creating hollow microcapsules of all-nanoparticle multilayers assembled at the two conditions mentioned above. Hollow microcapsules comprising  $TiO<sub>2</sub>$  and  $SiO<sub>2</sub>$  nanoparticles were obtained by calcinating the PS microspheres after the deposition of five bilayers of  $TiO<sub>2</sub>$  and  $SiO<sub>2</sub>$  nanoparticles at the two conditions. Figure 5 shows that, depending on the assembly conditions, two completely different microcapsules can be prepared. The calcinated hollow microcapsules (as evidenced by a broken capsule in the inset of Figure 5a) assembled at  $(TiO<sub>2</sub> 3.0/$  $SiO<sub>2</sub>$  3.0) show that because of the thick capsule walls, the microcapsules do not undergo the collapse that occurs during

<sup>(40)</sup> The values following TiO<sub>2</sub> and SiO<sub>2</sub> represent the pH of respective nanoparticle suspensions. For example,  $(TiO<sub>2</sub> 3.0/SiO<sub>2</sub> 5.0)$  represents multilayers assembled with  $TiO<sub>2</sub>$  and  $SiO<sub>2</sub>$  nanoparticle suspensions adjusted to pH 3.0 and 5.0, respectively.

<sup>(41)</sup> Netz, R. R.; Joanny, J. F. *Macromolecules* **<sup>1999</sup>**, *<sup>32</sup>*, 9013-9025. (42) Lvov, Y.; Decher, G.; Sukhorukov, G. *Macromolecules* **1993**, *26*,

<sup>5396–5399.&</sup>lt;br>Abu-Sharkh (43) Abu-Sharkh, B. *Langmuir* **<sup>2006</sup>**, *<sup>22</sup>*, 3028-3034.



**Figure 5.** SEM images of (a) (TiO<sub>2</sub> 3.0/SiO<sub>2</sub> 3.0)<sub>5</sub> and (b) (TiO<sub>2</sub> 3.0/SiO<sub>2</sub> 5.0)<sub>5</sub> hollow microcapsules. Samples prepared by depositing five bilayers of TiO<sub>2</sub> and  $SiO<sub>2</sub>$  nanoparticles on top of (PAH/PSS)<sub>5</sub>-coated PS microspheres were calcinated at 550 °C for 2 h. Due to the uneven evaporation of Au/Pd coatings for SEM imaging, the image on the top shows incomplete coating of Au/Pd on the TiO<sub>2</sub>/SiO<sub>2</sub> hollow microcapsules. Insets show high-magnification images of hollow microcapsules.

calcination of microcapsules assembled at  $(TiO<sub>2</sub> 3.0/SiO<sub>2</sub>)$ 5.0). The collapsed microcapsules (Figure 5b) indicate that because of the incomplete charge reversal, these capsule walls are extremely thin, even after the five cycles of the deposition process. Caruso et al. have shown that inorganic hollow spheres (without structural collapse) can also be created by calcinating polystyrene microspheres coated with charged inorganic nanoparticles and an oppositely charged polyelectrolyte.8

All-nanoparticle thin films comprising  $TiO<sub>2</sub>$  and  $SiO<sub>2</sub>$ nanoparticles have self-cleaning properties due to the presence of  $TiO<sub>2</sub>$  nanoparticles. Also, these films are porous in nature because of the presence of interstitial void volume between randomly packed nanoparticles.<sup>25</sup> The control of the porosity of these films will enable, for example, the investigation of effect of porosity on wetting behavior or on mechanical properties<sup>44</sup> of these all-nanoparticle thin films. Also, the ability to control chemical composition by controlling the assembly conditions of these all-nanoparticle multilayers would offer great advantages in creating functional thin films for specific applications. For example, Nakajima et al. have shown that a precise control over the amount of  $TiO<sub>2</sub>$  is crucial in maintaining the long-term stability of the superhydrophobicity in self-cleaning transparent superhydrophobic thin films.<sup>26</sup> We have examined the effect of assembly conditions on chemical composition, porosity, and refractive index of the  $TiO<sub>2</sub>/SiO<sub>2</sub>$  nanoparticle multilayers using a method based on ellipsometry described briefly in the Experimental Section and in detail in ref 25. This method allows the determination of porosity and chemical composition without making any assumption about the refractive index of each nanoparticle.

Figure 6a shows that the  $TiO<sub>2</sub>$  content in these allnanoparticle multilayers can be varied from 5 to 25 wt % or 2 to 10 vol % (see the Supporting Information). Within pH ranges where the average bilayer thickness is comparable to the size of the nanoparticles (i.e., pH range between 3.0 and 4.0 for both nanoparticle suspensions), the  $TiO<sub>2</sub>$  content could be varied from ca. 5 to 20 wt %. In general, increasing the pH of each nanoparticle suspension increased the amount of  $TiO<sub>2</sub>$  nanoparticles in the multilayers. An understanding of the effect of assembly conditions on chemical composition of the multilayers enables the fabrication of all-nanoparticle



**Figure 6.** (a) Weight fraction of TiO<sub>2</sub> nanoparticles, (b) porosity, and (c) effective refractive index of the multilayers as a function of assembly pH as determined via ellipsometry. TiO<sub>2</sub>/SiO<sub>2</sub> nanoparticle multilayer thin films with 12 bilayers were assembled on Si substrates for the determination of porosity and chemical composition.

thin films with desirable amount of  $TiO<sub>2</sub>$  nanoparticles. It should be possible, for example, to create self-cleaning thin films with varying degrees of photocatalytic activities.

<sup>(44)</sup> Stafford, C. M.; Harrison, C.; Beers, K. L.; Karim, A.; Amis, E. J.; Vanlandingham, M. R.; Kim, H. C.; Volksen, W.; Miller, R. D.; Simonyi, E. E. *Nat. Mater.* **<sup>2004</sup>**, *<sup>3</sup>*, 545-550.

The porosity of these films can also be varied as much as 10% depending on the assembly conditions, as seen in Figure 6b. It is noteworthy that the assembly pH that leads to thick bilayer growth results in lower porosity. This result indicates that the conditions that favor thick bilayer growth also favor the adsorption of densely packed nanoparticle layers. In addition to the control of assembly conditions, our previous study has shown that the use of nanoparticles of different sizes provides another approach to control the porosity of these all-nanoparticle thin films.<sup>25</sup>

Figure 6c shows that although changes in the refractive index of the films are observed, the difference in the refractive index between different conditions is very small (less than 0.05), even though the average bilayer thickness varies considerably (Figure 2). The lack of significant changes in the refractive index of the films can be explained by the fact that multilayer assembly conditions that lead to higher porosity, in general, result in greater volume fraction of  $TiO<sub>2</sub>$  nanoparticles in the multilayer thin films. As  $TiO<sub>2</sub>$ nanoparticles have a high refractive index of ∼2.3, a greater volume fraction of  $TiO<sub>2</sub>$  nanoparticles will offset the effect of a higher porosity; this self-compensatory behavior causes the refractive indices of the all-nanoparticle films studied here to remain essentially constant.

## **Conclusions**

We demonstrated that by controlling the assembly conditions (i.e., pH of each nanoparticle suspension) of multilayers comprising oppositely charged  $TiO<sub>2</sub>$  and  $SiO<sub>2</sub>$  nanoparticles, it is possible to vary the average bilayer thickness of multilayers as well as the porosity and chemical composition. Changes in the zeta-potential of  $TiO<sub>2</sub>/SiO<sub>2</sub>$  multilayers assembled on polystyrene microspheres at different conditions indicated that although complete charge reversal after each nanoparticle adsorption step leads to the growth of thick layers, incomplete charge reversal results in very little incremental growth. The chemical composition and porosity of these all-nanoparticles also could be varied by controlling the assembly conditions. These observations will facilitate the fabrication of all-nanoparticle thin films with characteristics to suit specific applications. The refractive index of the  $TiO<sub>2</sub>/SiO<sub>2</sub>$  nanoparticles multilayers showed negligible changes within the assembly conditions employed.

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**Supporting Information Available:** Volume fraction of TiO2 nanoparticles in TiO<sub>2</sub>/SiO<sub>2</sub> nanoparticle multilayers as a function of assembly pH. This material is available free of charge from http://pubs.acs.org/.

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